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# Sb<sub>2</sub>S<sub>3</sub> as a low-loss phase-change material for mid-IR photonics

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**Abstract:** We study the mid-IR optical properties of Sb<sub>2</sub>S<sub>3</sub> – a phase-change material (PCM) with high potential for mid-IR integrated photonics. We show that Sb<sub>2</sub>S<sub>3</sub> exhibits a high refractive index contrast  $\Delta n \approx 1$  between its two phases and an extinction coefficient  $k$  below  $10^{-2}$  at mid-IR wavelengths (up to  $\sim 10 \mu\text{m}$ ). We study the impact of 100 nm Sb<sub>2</sub>S<sub>3</sub> cladding on SiGe-on-Si waveguides and experimentally show that it brings additional propagation loss below 1 dB/cm in the 3.3–3.9  $\mu\text{m}$  wavelength range in both phases. These findings pave the way for new uses of Sb<sub>2</sub>S<sub>3</sub> in mid-IR integrated photonics.

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## 1. Introduction

Mid-IR wavelength range (commonly defined as spanning from 3 to 13  $\mu\text{m}$ ) covers the molecular absorption region of various atmospheric gases. Consequently, mid-IR integrated photonics, i.e. the integration of complex and advanced optical functionalities onto a chip, represents a promising avenue for the development of compact, cost-effective instruments for spectroscopy-based gas detection [1–6]. These structures are commonly fabricated with lithographic techniques which limit the reconfigurability and tunability of the resulting devices. Some trimming post-process capabilities were demonstrated by coating an additional layer on top of dielectric waveguides [7]. To go even further and enable a genuine post-fabrication tuning mechanism for these structures, one attractive approach is to integrate them in a hybrid manner with phase-change materials (PCMs). These materials can be reversibly switched between amorphous and crystalline phases exhibiting distinct optical properties.

Well-known examples of conventional PCMs are Ge<sub>2</sub>Sb<sub>2</sub>Te<sub>5</sub> (GST) [8,9] and VO<sub>2</sub> [10–14]. GST has been attracting intense attention due to its exceptional features, including a high refractive index contrast in the near-IR range between its two phases ( $\Delta n > 2.5$ ), low switching temperature ( $\sim 180^\circ\text{C}$ ) and its ability to maintain its state without any power supply. Numerous integrated devices operating around the telecom C-band such as optical memories [15], mode converters [16], reflection modulators [17], ring resonators [18], narrow-band filters [19] or phase shifters [20] based on GST were demonstrated. However, despite continuous studies and efforts to enhance its potential, its usability is still limited mostly to applications demanding light

amplitude modulation [21,22]. GST in crystalline phase shows sizeable extinction coefficient throughout the mid-IR range, resulting in substantial material absorption [19]. It is therefore not suitable for low loss applications. A successful integration of PCMs with low-loss mid-IR photonics could open the door to a new realm of applications and possibilities but until now has not been demonstrated.

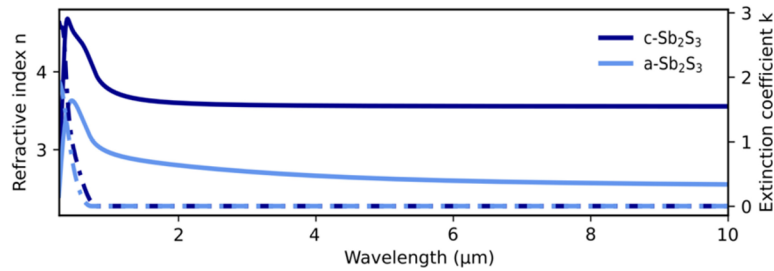
Recently, new attractive PCMs emerged such as  $\text{Ge}_2\text{Sb}_2\text{Se}_4\text{Te}$  (GSST) [23–25],  $\text{Ge}_2\text{Sb}_2\text{Te}_3\text{S}_2$  (GSTS) [26],  $\text{Sb}_2\text{S}_3$  or  $\text{Sb}_2\text{Se}_3$  [27–29] with the potential to be employed in low-loss devices. Particular interest was dedicated to  $\text{Sb}_2\text{S}_3$  as it manifests significantly higher figure of merit (defined as  $\Delta n/k_c$  where  $k_c$  stands for the extinction coefficient in the crystalline state) than GST while keeping similar phase switching time in the order of nanoseconds. Its potential was illustrated in image displays [30,31] and various integrated tunable devices [32–34]. Nonetheless, all demonstrations so far are limited to devices operating in the visible or near-IR wavelength range (i.e. below 2  $\mu\text{m}$ ). Until now, optical properties of  $\text{Sb}_2\text{S}_3$  in terms of refractive index contrast and propagation loss in the mid-IR have not been studied. Furthermore, the thickness of the  $\text{Sb}_2\text{S}_3$  layer in low-loss devices typically does not exceed a few tens of nanometers, thereby severely constraining the range of tunability. Optical properties of  $\text{Sb}_2\text{S}_3$  layers of different thickness were studied up to 2.5  $\mu\text{m}$  in Ref. [35] but no device operating with  $\text{Sb}_2\text{S}_3$  claddings thicker than 100 nm was demonstrated.

In this paper, we present the first studies of  $\text{Sb}_2\text{S}_3$  optical properties in the mid-IR range and examine its suitability for low-loss integrated tunable photonic devices. We perform spectroscopic ellipsometry measurements in both amorphous and crystalline phases for a broad wavelength range, between 560 nm and 10  $\mu\text{m}$ , i.e. deep into the mid-IR band. Next, we integrate this PCM onto mid-IR waveguides, by depositing a 100 nm thick  $\text{Sb}_2\text{S}_3$  cladding on top of SiGe-on-Si waveguides. By measuring their optical transmission in the mid-IR (3.3 to 3.9  $\mu\text{m}$ ) before and after crystallizing the  $\text{Sb}_2\text{S}_3$  material, we demonstrate that a PCM cladding in both phases introduces relatively low additional loss. This shows that  $\text{Sb}_2\text{S}_3$  is a highly promising PCM for tunable mid-IR photonic applications.

## 2. Ellipsometry measurements on a $\text{Sb}_2\text{S}_3$ /Si wafer

First, we study the properties of  $\text{Sb}_2\text{S}_3$  onto bare Si wafers. An amorphous  $\text{Sb}_2\text{S}_3$  layer was deposited using electron-beam physical vapor deposition (EBPVD) at room temperature with a deposition rate  $\sim 0.5 \text{ \AA/s}$ , an initial pressure of  $1.5 \times 10^{-6}$  mbar and an emission current of 5 mA. To facilitate the ellipsometry results analysis, the structure was not capped. Thicknesses of the PCM layer before and after crystallisation were first examined by X-ray reflectometry (XRR) and found to be 101.4 nm and 90.6 nm respectively. Mid-IR optical properties were studied using the IMPACT OPTICS Ellipsometry module. For the purpose of avoiding surface oxidation, sample's crystallization was performed under vacuum, with a partial pressure of  $\text{N}_2$  (1 mbar), in-situ in the ellipsometer chamber by heating up to the temperature of 270°C with a ramp rate of 10°C/min. Ellipsometry response of the sample at an incident angle of 70° was measured in the 350 nm–2  $\mu\text{m}$  range with a Uvisel2TM and in the 2  $\mu\text{m}$ –10  $\mu\text{m}$  range with a MIR polarimeter. Refractive index change in amorphous and crystalline  $\text{Sb}_2\text{S}_3$ , as extracted from the fits, is shown in Fig. 1. Details of the XRR and ellipsometry measurements can be found in the Supplement 1.

Numerical values for different wavelengths are presented in Table 1. The refractive index is always higher in the crystalline phase. The refractive index contrast  $\Delta n$  between crystalline and amorphous  $\text{Sb}_2\text{S}_3$  varies between 0.80 ( $\lambda=2.0 \mu\text{m}$ ) and 1.00 ( $\lambda=10 \mu\text{m}$ ). The extinction coefficient is found to vanish below the detection threshold ( $k < 10^{-2}$ ) of the ellipsometer for wavelengths beyond 815 nm for both phases.



**Fig. 1.**  $\text{Sb}_2\text{S}_3$  on Si wafer refractive index  $n$  (solid lines) and extinction coefficient  $k$  (dashed lines) ellipsometry measurements results in both phases from 260 nm to 10  $\mu\text{m}$ .

**Table 1.**  $\text{Sb}_2\text{S}_3$  refractive index  $n$  in amorphous and crystalline phases for selected wavelengths.

Wavelength	2 $\mu\text{m}$	4 $\mu\text{m}$	6 $\mu\text{m}$	8 $\mu\text{m}$	10 $\mu\text{m}$
$n$ (amorphous)	2.80	2.66	2.60	2.57	2.55
$n$ (crystalline)	3.60	3.56	3.56	3.56	3.55
$\Delta n$	0.80	0.90	0.94	0.99	1.00

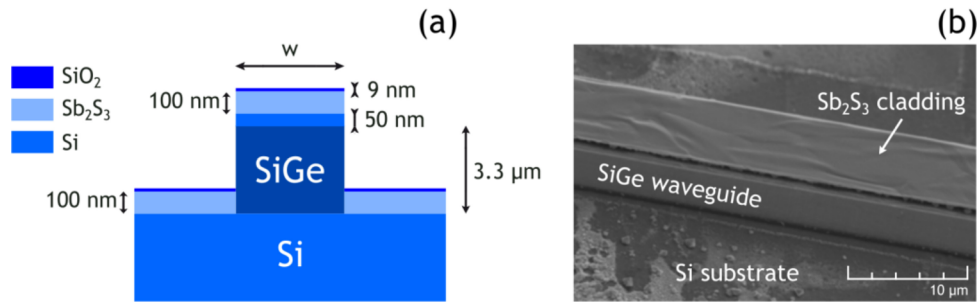
### 3. Optical characterisation of $\text{Sb}_2\text{S}_3/\text{SiGe}/\text{Si}$ waveguides

In the next part, we integrated a  $\text{Sb}_2\text{S}_3$  cladding layer onto a photonic chip containing SiGe-on-Si waveguides. By performing cut-back measurement, we studied the influence of PCM layer in both phases on waveguides propagation loss.

#### 3.1. Photonic chip description and fabrication

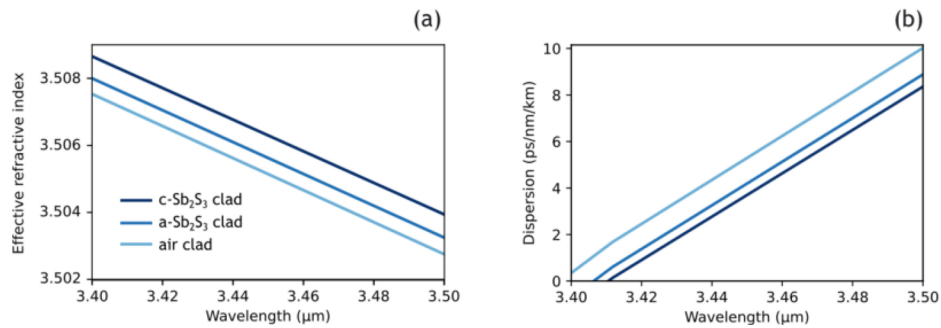
Photonic chip was manufactured on a 200 mm CMOS pilot line in CEA-LETI Grenoble. In the fabrication process, a 3.3  $\mu\text{m}$  thick  $\text{Si}_{0.6}\text{Ge}_{0.4}$  layer was grown on top of a Si substrate by Chemical Vapour Deposition (CVD) and encapsulated by a 550 nm thick Si layer. The surface roughness was removed by chemical mechanical polishing, resulting in a 50 nm thick Si encapsulation layer. Waveguides were then patterned by deep UV photolithography (DUV) followed by reactive ion etching (RIE). A Ge concentration of 40% was chosen for optimal trade-off between high nonlinear properties, transparency window and low propagation loss. Detailed fabrication procedures for the photonic chip can be found in the supplementary information of Ref. [36]. A 100 nm thick  $\text{Sb}_2\text{S}_3$  layer was deposited following the same procedure as described in Section 2. To avoid  $\text{Sb}_2\text{S}_3$  surface oxidization that would result in material's properties degradation, the sample was covered by EBPVD with a 9 nm thick  $\text{SiO}_2$  capping layer. A schematic of the waveguide cross section and a scanning electron microscope (SEM) picture after the crystallization are presented in Fig. 2. As is evident on the SEM image, the deposition of the  $\text{Sb}_2\text{S}_3$  layer exclusively occurs atop the waveguides, leaving sidewalls uncovered.

Based on the refractive index wavelength dependence presented in Section 2, we simulated the impact of a 100 nm  $\text{Sb}_2\text{S}_3$  cladding on the waveguide properties in the mid-IR range. Figures 3(a) and 3(b) show the simulated waveguide mode effective refractive index and dispersion as a function of wavelength respectively. Figure 3 results reveal that  $\text{Sb}_2\text{S}_3$  phase transition induces an effective refractive index and dispersion change of the order of 0.001 and 1 ps/nm/km, respectively. The limited effect can be attributed to the small confinement factor (defined in the Supplement 1) in the PCM, which does not exceed 0.14% as indicated in the intensity profile simulations presented in Fig. 4. For comparison, the overlap with the  $\text{SiO}_2$  capping is ten times less (0.012%). We highlight that this is not an intrinsic limitation, and this could be further

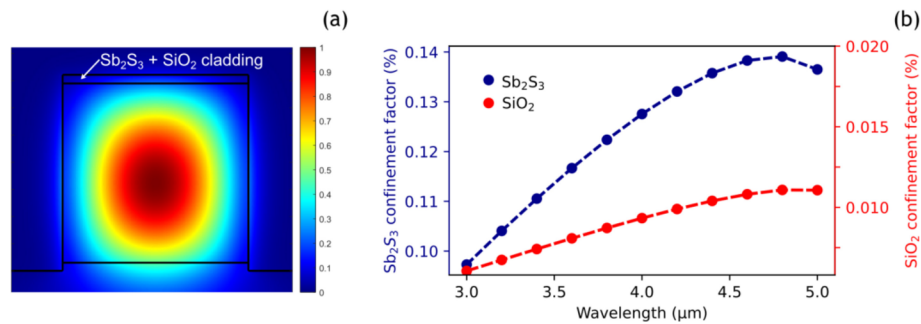


**Fig. 2.** (a) Cross section schematic and (b) tilted SEM picture with a 5 kV accelerating voltage of a  $\text{Si}_{0.6}\text{Ge}_{0.4}$  waveguide after crystallization.

enhanced by increasing the  $\text{Sb}_2\text{S}_3$  thickness or changing the waveguide geometry to increase the field overlap with the PCM layer.



**Fig. 3.** Simulated (a) effective refractive index and (b) dispersion profile of the TE<sub>0</sub> mode of the structure shown in Fig. 2(a) as a function of wavelength in the 3.40 to 3.50  $\mu\text{m}$  range for air, a- $\text{Sb}_2\text{S}_3$  and c- $\text{Sb}_2\text{S}_3$  cladded  $\text{Si}_{0.6}\text{Ge}_{0.4}$  waveguides. Simulations were performed with the use of Lumerical mode solver.



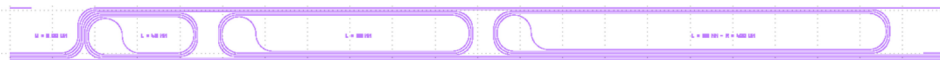
**Fig. 4.** Simulated (a) mode intensity profile (in the linear scale) of the structure for the wavelength of 3.0  $\mu\text{m}$  and (b) calculated confined factor in the  $\text{Sb}_2\text{S}_3$  (blue) and  $\text{SiO}_2$  (blue) layer for the wavelength range from 3.0 to 5.0  $\mu\text{m}$ .

### 3.2. Propagation loss measurements

Through numerical simulations, we find the theoretical upper limit of propagation loss in a waveguide-based geometry presented in Fig. 2(a) due to  $\text{Sb}_2\text{S}_3$  absorption as  $\sim 2$  dB/cm (associated

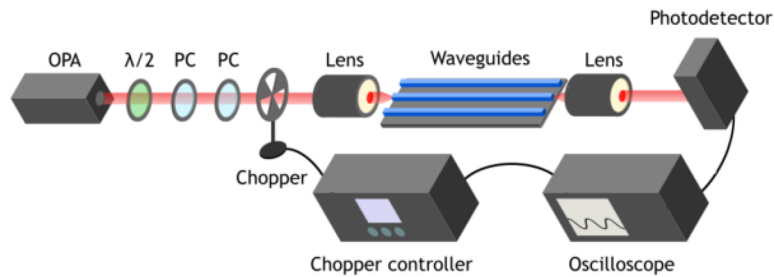
with extinction coefficient  $k$  value of  $10^{-2}$ ). In order to examine the PCM cladding's influence on waveguides below the ellipsometry detection limit, we tested additional propagation loss of the PCM covered waveguides using the cut-back method on three spiral waveguides with different lengths (4.2, 6.2, and 8.2 cm, see Fig. 5). Waveguide's width and height were  $3.0\ \mu\text{m}$  and  $3.3\ \mu\text{m}$ , respectively.

Our measurement setup, shown in Fig. 6, consisted of a laser source, a half-wave plate ( $\lambda/2$ ), two polarization controllers (PC) used for light attenuation, a chopper, two chalcogenide lenses, a PbSe photodetector and an oscilloscope. We used a tunable pulsed optical parametric amplifier (OPA, MIROPA-fs, Hotlight Systems) with a repetition rate of 63 MHz and  $\sim 200$  fs pulse duration as the light source. Measurements were performed under relatively low average power ( $\sim 1$  mW, corresponding to the pulse peak power of 50 W) to avoid the excitation of non-linear effects and modification of the PCM properties. For wavelengths varying from  $3.3$  to  $3.9\ \mu\text{m}$ , we coupled light in the TE polarization (due to the greater sensitivity of this mode to the presence of the PCM layer) to each of the spirals and measured the output signal with a Thorlabs PDA20H-EC photodetector.



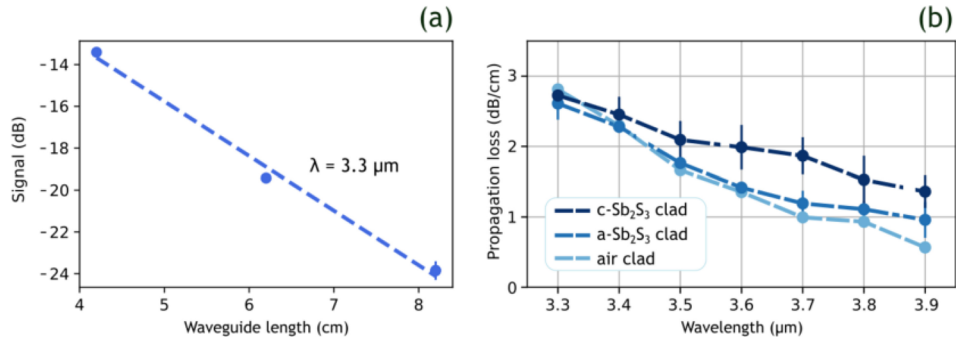
**Fig. 5.** Mask of spirals of different length used for propagation loss measurements.

Losses were extracted from the slope of the linear fit to the decimal logarithm of the output signal (see Fig. 7(a)). Facet and bending losses were assumed to be the same in each spiral (due to equal bending radii) so that their presence did not influence the slope. The sample's crystallization was performed by heating up to a temperature of  $270^\circ\text{C}$  under ambient pressure.

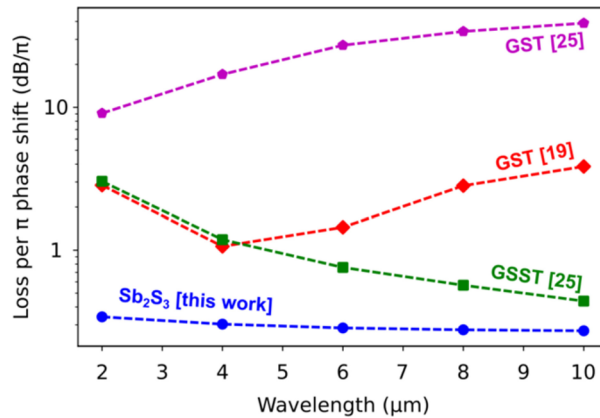


**Fig. 6.** Optical setup used for propagation loss measurements.

Results for air-, a-Sb<sub>2</sub>S<sub>3</sub>- and c-Sb<sub>2</sub>S<sub>3</sub>-cladded waveguides are presented in Fig. 7(b). We observe that propagation loss slightly decreases with increasing wavelengths for the three waveguide configurations, reaching around 1 dB/cm at  $\lambda = 3.9\ \mu\text{m}$ . While the air- and a-Sb<sub>2</sub>S<sub>3</sub>-cladded waveguides yield similar loss, there is a moderate increase of propagation loss for the c-Sb<sub>2</sub>S<sub>3</sub>-cladded waveguide with respect to the other two.



**Fig. 7.** (a) Linear fit to the output voltage signal in reference to the input signal as a function of spiral's length for  $3.0 \times 3.3 \mu\text{m}$   $\text{Si}_{0.6}\text{Ge}_{0.4}$  waveguides at a wavelength of  $3.3 \mu\text{m}$  for an a- $\text{Sb}_2\text{S}_3$  cladding and (b) Propagation losses at various wavelengths for air, a- $\text{Sb}_2\text{S}_3$  and c- $\text{Sb}_2\text{S}_3$  cladded SiGe waveguides, respectively. Error bars were determined using the uncertainty associated with the least squares fit method. The dashed lines are guides for the eye.



**Fig. 8.** Loss per  $\pi$  phase shift for different PCMs in the mid-IR range inferred based on ellipsometry results. Data are taken from [19] (GST red) and [25] (GSST and GST magenta). The significantly different performances of GST might stem from distinct stoichiometry or density attributed to different deposition techniques.



#### 4. Discussion

Propagation loss experienced by light propagating inside a PCM over a distance required to achieve a  $\pi$  phase shift can serve as a quantitative criterion for comparing materials in terms of their suitability for low-loss reconfigurable photonics. This can be calculated from the ellipsometry measurements of Section 2 via [20]:

$$Loss \left( \frac{dB}{\pi} \right) = \frac{20\pi}{\ln 10} \frac{k_c}{\Delta n} \quad (1)$$

With a refractive index contrast  $\Delta n$  between both phases approaching unity and an exceedingly low extinction coefficient in its crystalline phase ( $k_c < 10^{-2}$ ),  $\text{Sb}_2\text{S}_3$  exhibits loss per  $\pi$  phase shift of less than 0.3 dB beyond the wavelength of 3  $\mu\text{m}$  which positions it as a leading candidate for mid-IR photonics. A comparison of loss per  $\pi$  phase shift for different PCMs in the mid-IR range is presented in Fig. 8.

Regarding the hybrid waveguide results, Fig. 7 measurements show that, for wavelengths between 3.3  $\mu\text{m}$  and 3.9  $\mu\text{m}$ , the presence of a 100 nm thick  $\text{Sb}_2\text{S}_3$  cladding results in nearly negligible additional propagation loss in the amorphous phase, with a modest increase after crystallization. In both phases, the additional loss of the hybrid waveguide thus remains below 1 dB/cm. Considering the equally low (<0.3 dB) loss per  $\pi$  phase shift value, which more fairly accounts for the light interaction with the PCM,  $\text{Sb}_2\text{S}_3$  can be deemed highly promising for a variety of low-loss devices as it represents a noteworthy combination of low propagation loss and decent tunability capability. We note that SiGe-on-Si waveguides have been shown to exhibit propagation loss as low as below 0.2 dB/cm for wavelengths beyond 3.3  $\mu\text{m}$  [37]. The slightly higher absolute loss level presented in Fig. 7(b) for air-cladded waveguides may here originate from mechanical damages to the photonic chip during sample manipulation prior to the PCM deposition. It is however not an intrinsic limitation of this platform. Our results show that both contributions to loss (absorption and scattering) remain low in a c- $\text{Sb}_2\text{S}_3$  waveguide-based geometry, as would be used in practical device implementation. Since the extra propagation loss for the waveguide with  $\text{Sb}_2\text{S}_3$  in the crystalline phase is lower than the theoretical upper limit predicted by simulations (2 dB/cm) considering the most pessimistic extinction coefficient of  $10^{-2}$  (see Section 2), the current study does not allow us to identify its origin. It could stem from  $\text{Sb}_2\text{S}_3$  residual absorption or light scattering at grain boundaries and at the  $\text{Sb}_2\text{S}_3/\text{SiGe}$  interface. A prospective avenue may involve the systematic analysis of the influence of diverse fabrication parameters such as deposition rate, capping layer thickness, crystallization temperature, pressure and so on. We next plan to test waveguides with thicker  $\text{Sb}_2\text{S}_3$  claddings. Such configurations would be useful to increase the tunability of waveguide properties upon PCM phase changes.

These findings lead to a further extension of the operational wavelength range of tunable devices such as phase shifters or mode converters to the mid-IR. The possibility of depositing PCM claddings with thicknesses exceeding 100 nm without introducing substantial amount of additional propagation loss should lead to an enhancement of the tunability of waveguide's properties such as dispersion profile and effective refractive index, therefore significantly improving the versatility of integrated devices, including ring resonators or directional couplers. The findings can also pave the way for an entire new group of applications such as waveguide's dispersion profile modification for nonlinear light sources.

#### 5. Conclusion

We have measured for the first time the optical properties of  $\text{Sb}_2\text{S}_3$  thin films in the mid-IR wavelength range and demonstrated their suitability for low-loss integrated photonics in this wavelength range. Thanks to ellipsometry measurements, we showed that the refractive index contrast between a- $\text{Sb}_2\text{S}_3$  and c- $\text{Sb}_2\text{S}_3$  was high ( $\Delta n \approx 1$ ) up to 10  $\mu\text{m}$  wavelength and found



extinction coefficients below the detection threshold ( $k < 10^{-2}$ ) in the entire measurement range. We demonstrated that the deposition of 100 nm of  $\text{Sb}_2\text{S}_3$  on top of SiGe-on-Si waveguides resulted in additional propagation loss below 1 dB/cm regardless of the  $\text{Sb}_2\text{S}_3$  phase. These findings create opportunities for the use of  $\text{Sb}_2\text{S}_3$  in reconfigurable low-loss integrated mid-IR devices.

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**Disclosures.** The authors declare no conflicts of interest

**Data availability.** Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

**Supplemental document.** See [Supplement 1](#) for supporting content.

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